



**HAL**  
open science

# Atomic scale insights on the microstructure evolution of urania under irradiation

A. Chartier, L. Vanbrutzel

► **To cite this version:**

A. Chartier, L. Vanbrutzel. Atomic scale insights on the microstructure evolution of urania under irradiation. MMM 2012 - 6th International Conference on Multiscale Materials Modeling, Oct 2016, Dijon, France. cea-02442352

**HAL Id: cea-02442352**

**<https://cea.hal.science/cea-02442352>**

Submitted on 16 Jan 2020

**HAL** is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

# Atomic scale insights on the microstructure evolution of urania under irradiation



Alain Chartier and Laurent Van Brutzel  
CEA-Saclay, 91191 Gif-sur-Yvette Cedex, France

université  
PARIS-SACLAY

## Introduction

Urania is commonly used as a fuel in nuclear industry. Urania is heavily irradiated during its in-reactor stay, and faces drastic microstructural modifications, including few percents swelling and increase of dislocation density. Dislocations are identified as perfect dislocations loops and transform with increasing fluence into lines at sufficiently high temperature – i.e. at 600 °C. However, the early stages of their nucleation are hardly attainable experimentally. One commonly infers that their nucleation is related to the aggregation of point defects or defects clusters into dislocations. In the present paper, we clarify the first steps of the effect of irradiation on urania by combining molecular dynamics simulations and experimental investigations [1].

## Methodology

### Molecular dynamics simulation

- Morelon potentials
- NPT ensemble

### Irradiation

- Frenkel pairs creation every 2 ps → dose rate  $K_0 = 1.6 \cdot 10^9$  dpU/s
- recombination regime

### Analysis

- Dislocations with DXA algorithm in Ovito [2]
- Point defects with Voronoi
- Swelling extracted by indexing XRD peaks

## Comparison with experiments:

Experimental dose rate  $K_0 = 10^{-3}$  dpa/s

Characteristic distance after time  $\tau = 1$  s of irradiation →  $(V_{at}/(K_0 \times \tau))^{1/3} = 34$  Å  
Characteristic distance for self-diffusion →  $(D \times \tau)^{1/2}$

$D(U \text{ à } 900K) = 10^{-31} \text{ m}^2/\text{s}$  [3] →  $1.18 \cdot 10^{14}$  s for 34 Å length diffusion

→ recombination regime in experiments

Dislocation density  $\rho$  in recombination regime:  $\rho \propto K_0^{1/6}$  [4]

→ Rescaling of dislocation densities by a factor of  $\sim 10^2$

## Nucleation and growth of dislocations

### Stage 1 (up to 0.07 dpU marked with grey dashed line):

- Increase of point defects concentrations, with equal concentration of vacancies and interstitials (bottom of Figure) → as predicted by rate equations.
- Recombination lifetime of uranium Frenkel pairs = 1.6 ps according to the relation  $\tau_{FP} = 1/K_0 \tau_1^2$  → spontaneous recombination.

### Stage 2 (between 0.07 dpU and 0.25 dpU (marked with red dashed line):

- Frank loops nucleate ( $1/3\langle 111 \rangle$  middle of Figure) from point defect clustering.
- Interstitials decrease (bottom of Figure), indicating a dislocation bias for them. → Frank loops of interstitial character.

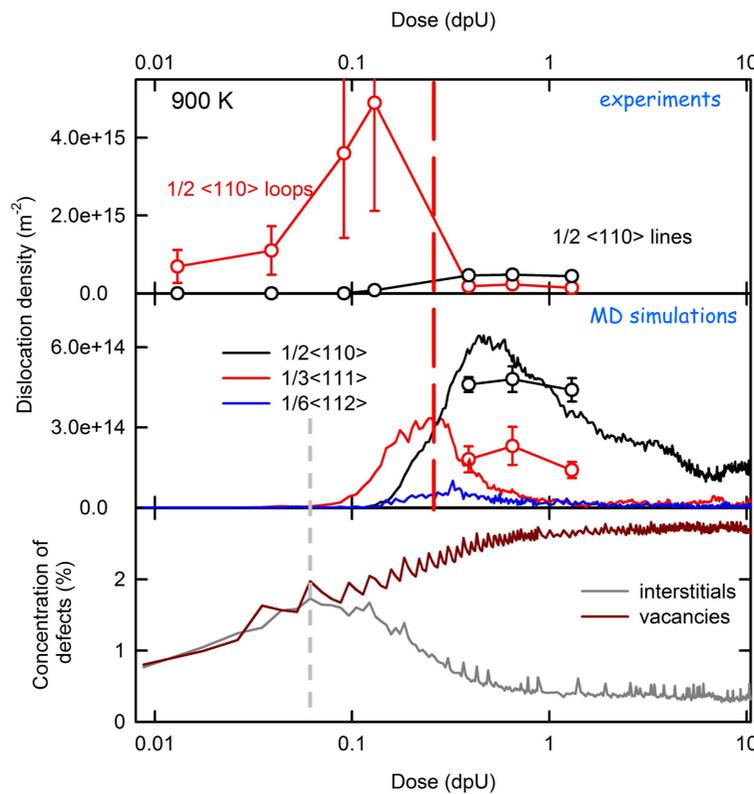
### Stage 3 (0.25 dpU up to 0.35 dpU):

- Frank loops transform into perfect loops ( $1/2\langle 110 \rangle$  middle of Figure)
- Mechanism suggested by Baufeld [5]:  $\frac{a}{6}\langle 11\bar{2} \rangle + \frac{a}{3}\langle 111 \rangle \rightarrow \frac{a}{2}\langle 110 \rangle$   
see Shockley partials ( $1/6\langle 112 \rangle$ ) and their connections with Frank and perfect loops (inset vertical Figure).

- Frank loops not seen experimentally due to their sub-nanometer size.
- Perfect loops observed in experiments and calculations.

### Stage 4 (above 0.35 dpU):

- Perfect loops convert into lines.
- Forest dislocations stabilize above 1 dpU, dislocation densities of  $5 \times 10^{14} \text{ m}^{-2}$ .
- Defect concentrations stabilize with vacancies / interstitials = 6  
3-4 estimated by Nakae *et al* [6] (bottom of Figure).



## Swelling

### Comparison with experiments:

- Lattice expansion close to experiments [7,8,9]
- Lattice expansion – contraction peak observed experimentally in nuclear fuel [10]

### Swelling deconvolution:

$$\frac{\Delta V}{V} = V_f^i \times \rho_{line} + \frac{R_{loop} \times b}{2} \times \rho_{loop} + \frac{V_{rel}^i}{\Omega} \times C_i + \frac{V_{rel}^v}{\Omega} \times C_v$$

- $V_{rel}^v$  and  $V_{rel}^i$ : relaxation volumes of vacancies and interstitials
- $\Omega$ : atomic volume
- $\frac{1}{2} R_{loop} \times b$ : formation volumes of dislocation loops
- $b$ : Burgers vector
- $R_{loop}$ : average loop radius
- $V_f^i$ : formation volume of lines
- $\rho_{line}$  and  $\rho_{loop}$ : line and loop dislocation densities
- $C_i$  and  $C_v$ : interstitial and vacancy concentrations

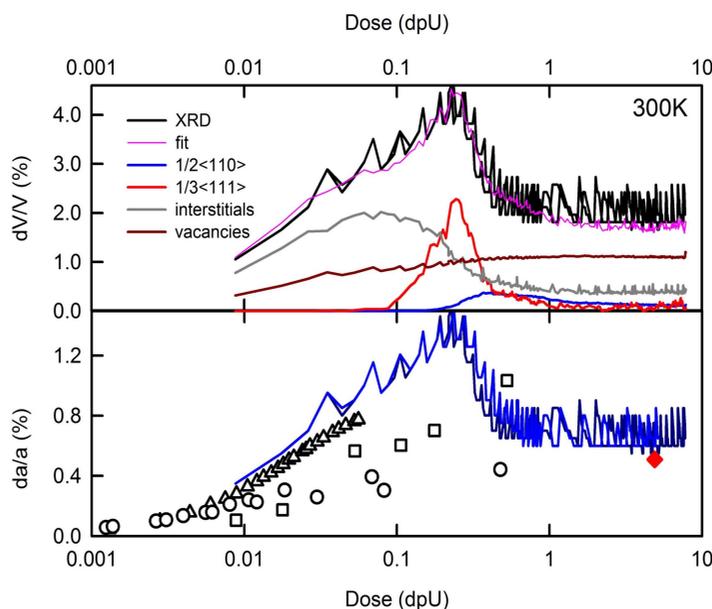
### Swelling stage

- 0.0 to 0.1 dpU → point defects
- 0.1 to 0.2 dpU → enhanced by Frank loops

### Contraction stage

- 0.2 to 1.0 dpU → transformation of Frank loops into unfaulted loops dislocations.

Residual swelling at steady state is mainly related to vacancies and to interstitials.



## Conclusion

Starting from a defectless urania, we observe the nucleation and growth of dislocations under Frenkel pairs accumulation.

Detailed analysis shows a four stages evolution:

- an increase of point defects
- then the nucleation of Frank loops  $1/3\langle 111 \rangle$  from the aggregation of point defects
- the transformation of Frank loops into perfect loops  $1/2\langle 110 \rangle$
- and finally their stabilization as lines.

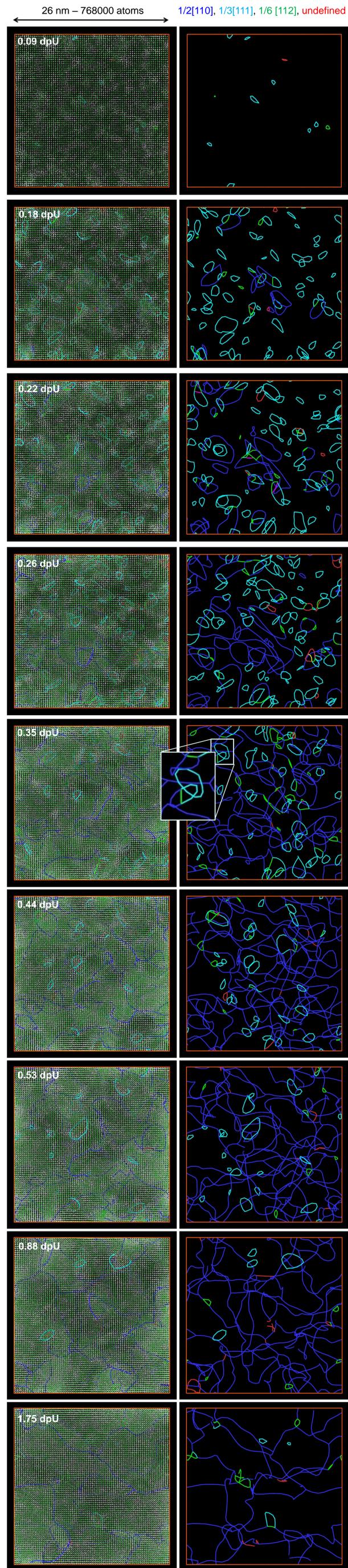
Our simulations also show a swelling up to 3.2% when point defects are present, this swelling suddenly decreases to 1.5%, as soon as dislocations nucleate.

Nucleation and growth of dislocations in UO<sub>2</sub> operate in the spontaneous recombination regime.

## References

- [1] A. Chartier, C. Onofri, L. Van Brutzel, C. Sabathier, O. Dorosh, and J. Jagielski, submitted
- [2] <http://www.ovito.org/>
- [3] E. Moore, C. Guéneau, and J.-P. Crocombette, J. Solid State Chem. 203 (2013) 145.
- [4] J. Rest, J. Nucl. Mater. 326 (2004) 175.
- [5] B. Baufeld, D. Baither, U. Messerschmidt, M. Bartsch, and I. Merkel, J. Am. Ceram. Soc. 76 (1993) 3163.

- [6] N. Nakae, Y. Iwata, and T. Kiriha, J. Nucl. Mater. 80 (1979) 314.
- [7] W.J. Weber, J. Nucl. Mater. 98 (1981) 206.
- [8] K. Hayashi, H. Kikuchi, and K. Fukuda, J. Nucl. Mater. 248 (1997) 191.
- [9] A. Debelle, A. Boule, F. Garrido, and L. Thomé, J. Mater. Sci. 46 (2011) 4683.
- [10] J. Spino, and D. Papaionnou, J. Nucl. Mater. 281 (2000) 146.



26 nm – 768000 atoms 1/2[110], 1/3[111], 1/6[112], undefined