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C$_{60}$ as an Atom Trap to Capture Co Adatoms

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ABSTRACT: C$_{60}$ molecules were used to trap Co adatoms and clusters on a Au(111) surface using atomic/molecular manipulation with a scanning tunneling microscope. Two manipulation pathways (successive integration of single Co atoms in one molecule or direct integration of a Co cluster) were found to efficiently allow the formation of complexes mixing a C$_{60}$ molecule with Co atoms. Scanning tunneling spectroscopy reveals the robustness of the π states of C$_{60}$ that are preserved after Co trapping. Scanning tunneling microscopy images and density functional theory calculations reveal that dissociated Co clusters of up to nine atoms can be formed at the molecule–substrate interface. These results open new perspectives in the interactions between metal adatoms and molecules, for applications in metal–organic devices.

INTRODUCTION

Building artificial nanostructures by atomic/molecular manipulation with a scanning tunneling microscope (STM) offers the opportunity to unravel the physical phenomena that govern the matter at the nanometer scale.$^{1,14}$ The rise of molecular electronics and spintronics has motivated a growing interest in molecule-metal interactions. STM experiments have been performed on individual manipulated molecule-metal conjugates,$^{5,6,13}$ providing a physical understanding of the interaction between molecules and metal atoms which is critical to build new generation molecular electronic devices and molecular machines. Being highly symmetrical, chemically reactive and produced easily in large amount, C$_{60}$ is one of the most important members in the nanocarbon family$^{15}$ and is a natural candidate for the study of model systems. The electronic spectrum and spatial mapping of molecular states of C$_{60}$ adsorbed on different metallic substrates have been investigated by STM/STS$^{16,17}$ and the interaction between a C$_{60}$ molecule and K adatoms has been measured using lateral manipulation and revealed a charge transfer from K atoms to C$_{60}$.$^{1}$ The interaction between C$_{60}$ molecules and transition metals has also attracted many attentions.$^{18–28}$ Significant magnetoresistance values have been reported for devices of C$_{60}$-based spin valves with the electrode of cobalt.$^{26,27}$ Furthermore, mixed thin films of Co and C$_{60}$ exhibited large magnetoresistance.$^{22,23}$ In such organometallic films, a charge transfer of 0.2 electrons/Co atom to the C$_{60}$ molecule was measured, and a saturation composition was found for Co$_3$C$_{60}$ above which Co dense clusters are formed.$^{29,30}$

In this work, using a low temperature STM, we manipulate atoms and molecules to build C$_{60}$--Co complexes and study their electronic properties by scanning tunneling spectroscopy (STS). We reach an x value of 9, which is beyond the above-mentioned saturation value of 5 in mixed films and is larger than previous manipulated metal-molecule complexes. We perform dI/dV spectroscopy and conductance mapping to measure the electronic properties of the complexes. We find that although the interaction between C$_{60}$ and Co is strong enough to obtain a C$_{60}$-Co$_x$ that can be moved on the surface as a single entity, the π states of C$_{60}$ are totally preserved; i.e., their energy position and wave function are not noticeably modified by the interaction with Co atoms and no evident charge transfer to these states is measured.

EXPERIMENTAL METHOD

The experiments were performed using a commercial Omicron low temperature STM operating under ultrahigh vacuum (UHV) ($<$10$^{-10}$ mbar) and at low temperature (5 K). We used a chemically etched tungsten tip for the STM measurement. The differential conductance (dI/dV) measurements were performed using the lock-in technique. The clean Au(111)/mica substrate was prepared by several cycles of Ar$^+$ sputtering (900 eV) followed by annealing at 600 K under UHV. The Co atoms were deposited by thermal evaporation of a 99.99% pure Co rod. C$_{60}$ molecules were deposited from a...
C₆₀ powder sublimated in a crucible heated at 600 K. These depositions were made in situ on the Au(111)/mica surface maintained at low temperature (5 K). Atomic and molecular manipulation were performed using lateral manipulation with the STM tip. In a typical molecular manipulation process, at first the tip is moved around the molecule and the scanning is stopped. Then the tip height is modified in order to reduce the tip–sample distance with the goal to increase the tip–molecule interaction. Then the tip is moved laterally along a straight line to the final position while the molecule moves under the influence of the tip. Finally the tip is retracted vertically to the initial height and the scanning is restarted. The molecule is then left at the final position on the surface. The tunneling resistance is used to estimate the tip–molecule distance and interaction strength. The larger the tunneling resistance is, the stronger is the tip–molecule interaction. In our experiment, molecular manipulation could be achieved either in constant current or in constant height mode. In the constant current mode the feedback loop was active during the manipulation with a typical tunnel resistance of 5 MΩ. In the constant height mode, the tip was first placed on top of the molecule (typical imaging parameter 1 V, 10 pA), then the feedback loop was turn off and the tip–sample distance reduced by about 5 Å before performing the manipulation with the tip maintained at constant height.

**RESULTS AND DISCUSSION**

After deposition of Co and C₆₀ on Au(111) at 5 K, single molecules and adatoms are clearly observed on the surface as shown in Figure 1a. Using the STM tip we performed molecular manipulation and moved a C₆₀ molecule toward a Co adatom, as shown in Figure 1a,b. In order to gain some insight into the cohesion of the C₆₀–Co ensemble we performed an additional manipulation step (indicated by the arrow in Figure 1b) and moved the molecule away from the initial position of the Co adatom. After this manipulation, the Co adatom is no more observed on the surface, as shown in Figure 1c, indicating that it has been displaced together with the C₆₀ and is now attached to the molecule. This molecule–atom ensemble will be called complex in the following. This C₆₀–Co complex was in turn successively moved toward individual adatoms, as shown in Figure 1b–f, in order to build C₆₀–Coₖ and explore the scalability of the process.

By this lateral manipulation 6 Co atoms were taken by one C₆₀ to build a C₆₀–Co₆ complex. Another way to build a C₆₀–Coₖ complex is to move a molecule toward a Coₖ cluster. To do this, we built a Coₖ cluster by atomic manipulation and then moved a C₆₀ molecule toward the cluster as shown in Figure 2a,b. Then we displaced away this molecule by an additional manipulation step from the initial position of the Coₖ cluster (Figure 2b,c). After this manipulation, the Coₖ cluster is no more observed on the surface it is attached to the molecule and in turn formed a C₆₀–Coₖ complex. In order to show the complex mobility and put more Co atoms into the complex, this C₆₀–Coₖ complex was moved successively by the STM tip toward the surrounding Co adatoms (Figure 2c). Finally, we have been able to add up to nine Co atoms to the C₆₀ molecule (Figure 2d). It has to be noted that it was more difficult to move the complex on the surface when more Co atoms were contained in the complex.

The methods described above are both efficient to obtain the C₆₀–Coₖ complex although the building pathway is very different. The structure of these complex is therefore an open question that we address in the following. We studied the apparent height of C₆₀ and C₆₀–Coₖ measured from the topographic STM image. Various apparent heights have been measured for C₆₀ organized on different metals. The apparent height of a C₆₀ molecule measured by STM depends on the orientation of the molecule, the STM tip state, the adsorption site on the surface, and electronic differences caused by surface interaction. Therefore, the height of a molecule cannot be explained by a purely geometrical model as electronic effects must be taken into account. The electronic effects will be discussed below and we first address the expected geometric height. Because of its atomic structure, a fulleren C₆₀ exhibits five typical orientations: (a) hexagon, (b) pentagon, (c) 6:6 bond, (d) 5:6 bond, and (e) apex atom, which correspond to the configurations of a hexagonal ring, a pentagonal ring, a bond joining two adjacent hexagonal rings, a bond joining a hexagonal and a pentagonal ring, and an apex atom riding on the surface, respectively. The geometric height (without considering the substrate) of pure C₆₀ with different orientations is in the range of 6.28–6.94 Å as given by an atomic model with a C–C bond length of 1.43 Å. A van der Waals distance of about 2.5 Å should be added to estimate the geometric height of an adsorbed molecule. Here, for C₆₀ molecule or C₆₀–Coₖ complex on Au(111) surface, the apparent height is measured as the vertical distance between the top of the molecule and the substrate along a scan line parallel to the reconstruction line (inset of Figure 3). As...
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Figure 2. Series of STM images obtained during successive manipulations of a C_{60} molecule toward a Co cluster and Co adatoms. (a) A group of 6 Co adatoms (marked by a rectangle) is manipulated to form a Co_{6} cluster (b). (b to c) A C_{60} molecule is moved onto this cluster to fabricate C_{60}−Co_{6}. Three neighbor Co adatoms are taken by C_{60}−Co_{6} and finally a C_{60}−Co_{9} complex is built (d). To guide the eyes, the arrows indicate the moving direction of the molecule. All the STM images are taken with a bias voltage and current set point of 2 V and 10 pA, respectively.

Figure 3. Apparent height (h) of C_{60} and C_{60}−Co_{x} measured by STM. The dotted line fits the average height at every x value. The black triangle and red circle symbols correspond to one by one and cluster manipulation, respectively. Inset: the profile of a C_{60} molecule. All the apparent height measurements were carried out from STM images taken with a bias voltage of 2 V.

illustrated in Figure 3, the apparent height of C_{60} is measured in the range of 4.9−7.3 Å. This dispersion of the apparent height can be due to the molecular orientation, the adsorption sites or more complex effects. We measured that for a given orientation the height of C_{60} on a given adsorption site (fcc or hcp) can vary by about 1.6 Å. Therefore, the effect of orientation or adsorption site on the apparent height can not be clearly identified. However, the measured height is clearly smaller than the expected geometric height which may be due to electronic effects. It is clear that the C_{60}−Co_{x} complexes also exhibit apparent height distributions. Despite this distribution, the average apparent height of C_{60}−Co complex increases when the number of Co atoms increases from 0 to 9 as shown in Figure 3. This is also another evidence for the formation of the complex as the variation of the height reveals a difference between each structure. However, this increase does not correspond to the sum of C_{60} and Co clusters taken separately. For example, comparing the apparent height of C_{60}−Co_{6} and pure C_{60}, there is an increase of ~2 Å, which is slightly larger than the height of a Co adatom (~1.6 Å) and much smaller than the apparent height of Co_{x} cluster (~5.2 Å). As will be discussed below, the electronic spectrum of the molecule is not strongly modified after the formation of a C_{60}−Co complex, and therefore we expect that electronic effects will not play a major role in the height variations that are measured. It is worth to note that the complexes fabricated by either one-by-one (black triangle in Figure 3) or cluster (red circle in Figure 3) manipulation show the same trend in the apparent height. It suggests that the two methods, one-by-one and cluster manipulation, lead to complexes with the same structure. This can be understood as the typical time constant of the manipulation process (typically 1 ms between two successive tip displacements during manipulation) is much larger than the typical time constant of atomic relaxation,^{38} therefore the system can be considered as being in equilibrium during the process and independent of the pathway followed to build the complex.

In order to investigate the electronic interaction between C_{60} and Co atoms, we measured the influence of complex formation on the molecular orbitals of C_{60}. We recorded dI/ dV spectra with the STM tip located above the center of C_{60} molecules and different C_{60}−Co_{x} complex obtained separately. As illustrated in Figure 4a, three main resonance peaks, corresponding to the highest occupied molecular-orbital (HOMO), the lowest unoccupied molecular-orbital (LUMO), and LUMO+1 states, respectively, are observed in the differential spectrum of C_{60} molecules. Similar features are found for the C_{60}−Co_{x} complex with slight shifts of peak position toward both high and low energies (Figure 4a). For the energy of the HOMO state, we determined the energy position after numerical derivation of the spectrum allowing us to measure the zero-slope point. Therefore, no charge transfer can be clearly evidenced between Co and C_{60}. As illustrated by Xinhua et al., there is no obvious dependence of peak position on molecular orientation of C_{60} on Au(111).^{17} The average energies of the HOMO, LUMO, and LUMO+1 states are around ~2.0, +1.1, and +2.4 eV as shown in Figure 4b for C_{60} molecule and C_{60}−Co_{x} complex.

Simultaneously to the topographic STM image, we measured conductance maps on C_{60} molecule and C_{60}−Co_{x} complex with different orientations. Such conductance maps allow to visualize the molecular orbitals at given bias voltage. As shown in Figure S1 in the Supporting Information, no clear difference is observed between the molecular orbitals of single C_{60} and those of complexes. This shows that although the interaction between C_{60} and Co is strong enough to capture single atoms or cluster, the influence of the hybridization between Co orbitals and C_{60} π states is rather weak, so that the molecular orbitals remain unchanged. If the Co atoms were on the top of or beside the C_{60} molecule, a difference should be observed on the conductance map. Considering the fact that the difference between the apparent height of C_{60}−Co_{x} and pure C_{60} is slightly larger than the apparent height of Co adatom and much

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In order to interpret the experimental results, we performed DFT (density functional theory) calculations using a plane wave electronic structure package Quantum-ESPRESSO. Local density approximation in Perdew and Zunger parametrization was used for exchange-correlation functionals. The plane-wave cut-offs were set to 30 and 300 Ry for the wave functions and for the charge density, respectively. A Marzari-Vanderbilt broadening scheme with a broadening width of 0.15 eV was used. The Au(111) substrate was simulated by a 5 layers Au slab with a bulk lattice parameter found to be 4.06 Å, close to experimental value of 4.08 Å. The first three Au layers were fixed while two other layers and C60 complex were relaxed until atomic forces become smaller than 0.001 eV/Å. A (4 × 4) in-plane periodicity was adopted for Co/Au(111), C60/Au(111), and C60−Co/Au(111) in order to avoid unphysical interactions between adsorbates and the 2D Brillouin Zone (BZ) was sampled with (4 × 4 × 1) k-points mesh. For a bigger C60−Co/Au(111) complex, a bigger (5 × 5) in-plane supercell was employed.

To provide a simple energetic reasoning for a possibility to drag a single Co atom by a C60 molecule we present in Table 1 binding energies of Co atom to free C60 molecule as well as to Au surface at two different adsorption sites: at "fcc-hollow" and "bridge". The difference between the two latters gives an estimate for the diffusion barrier of Co atoms along a Au surface, $E_{\text{diffusion}}=E_{\text{fcc-hollow}}-E_{\text{bridge}}$. This value is much smaller than that for Co/Au(111) and reflects thus a general result of suppression of the Co magnetism due to binding to C60. The Co atom will bind to C60 and slide with it along the Au surface, remaining, however, always in contact with the surface, exactly as it is found in experiment. Note that the spin magnetic moment for Co/C60 was found to be $m_S \approx 1 \mu_B$ which is twice smaller than that for Co/Au(111), $m_S \approx 2.4 \mu_B$.

Next, we have studied the geometry and electronic structure of C60/Au(111), C60−Co/Au(111), and C60−Co/Au(111) complexes (Figure 5). The lowest-energy configuration of C60/Au(111) corresponds to a C60 hexagon placed right above an hcp-hollow site of a Au surface, in a good agreement with previous DFT calculations. In the case of C60−Co/Au(111), the Co atom was found to bind at hollow site to a Au surface and at 6:6 bond to a C60 molecule, in agreement with our results for Co/Au(111) and Co/C60 presented above. The total spin moment of C60−Co/Au(111) was found to be around 1.42 $\mu_B$. This value is much smaller than that for Co/Au(111) (2.42 $\mu_B$) and reflects thus a general result of suppression of the Co magnetism due to binding to C60. For the C60−Co/Au(111) complex, we performed atomic relaxations with two different initial configurations: (i) Co atoms were distributed in some random fashion around C60 molecule (no any statistics was done since calculations are rather heavy); (ii) Co atoms form a compact 2D monolayer island on Au(111) with a C60 placed on top of it. We found that the first configuration is much more energetically favorable with respect to the second one, with an energy gain of about 0.97 eV. The Co atoms prefer to form a noncompact structure around C60, as shown on inset of Figure 5c, which points to a strong tendency to dissociation of Co clusters due to the interaction with the C60 molecule. The total spin moment for this big complex was found to be around 13.71 $\mu_B$ which corresponds roughly to 1.52 $\mu_B$ per Co atom.

We next check out the height change of C60 molecule due to the presence of Co adatoms on a Au surface, taking as a reference the topmost C60 atom of C60/Au(111). The measured heights were found to be increased by ~0.14 and ~0.75 Å for the C60−Co/Au(111) and C60−Co/Au(111)
could be probed separately by changing the direction of the spin-polarized tunneling diode.

The vacuum LDOS of the sample (the surface with adsorbate) were calculated at the tip position and at the energy corresponding to the applied voltage \( V \). In practice, the vacuum LDOS is integrated over a small cubic box of the size which we choose here to be 0.4 Å. In Figure 5 such vacuum LDOS are presented for all three molecular complexes and for two different lateral positions. One can see that the curves of LDOS correspond pretty well to the STS spectra shown in Figure 4. Their shape however depends somewhat on the lateral box position. In particular, LUMO+1 derived states (at ~1.55 eV, see discussion below) are almost unvisible for the box represented by dashed lines.

In order to identify various features in vacuum LDOS, we also show in Figure 5a the DOS of \( \text{C}_{60}/\text{Au}(111) \) projected onto different molecular orbitals of a free \( \text{C}_{60} \) molecule. This allows us to ascribe clearly the two LDOS peaks at ~0.55 eV and ~1.55 eV to LUMO and LUMO+1 derived states, respectively, and a small feature at ~1.25 eV (hardly visible for \( \text{C}_{60}−\text{Co}/\text{Au}(111) \)) to HOMO orbitals. Interestingly, a broad feature at \( E < ~1.5 \text{ eV} \), seen in all LDOS curves, originates from the Au surface states, since, as we have checked, the similar structure was also observed for a clean Au(111) surface (without adsorbates). It should be noted however that the DFT is generally known to underestimate a LUMO–HOMO gap which was found here to be \( \approx 1.8 \text{ eV} \) while the value of \( \approx 2.6 \text{ eV} \) can be extracted from experimental STS spectra (see Figure 4).

Overall, no significant change in vacuum LDOS curves was found among \( \text{C}_{60}/\text{Au}(111) \), \( \text{C}_{60}−\text{Co}/\text{Au}(111) \) and \( \text{C}_{60}−\text{Co}/\text{Au}(111) \) complexes, which confirms the experimental results on STS spectra (Figure 4). Finally, we note that in Figure 5 the total LDOS, \( n_\uparrow = n_\uparrow^1 + n_\uparrow^2 \), is presented which simulates the STS spectra taken with nonmagnetic STM tip. If the tip were spin-polarized it would select a preferable spin channel so that \( n_\uparrow^1 \) could be probed separately by changing the direction of the tip polarization. We did not find, however, significant spin polarization of vacuum LDOS (Figure S2 in the Supporting Information).

**CONCLUSIONS**

Through atomic and molecular manipulation of \( \text{C}_{60} \) molecules and Co adatoms on clean Au(111) surface at low temperature (5 K) using scanning tunneling microscopy, the \( \text{C}_{60}−\text{Co}_x \) complex can be formed in two different ways: the molecule is successively moved toward individual adatoms, or a Co cluster is built by atomic manipulation then the molecule is moved toward the cluster. We have been able to add up to nine Co atoms to a \( \text{C}_{60} \) molecule, which is, to our knowledge, the largest number achieved in such molecule-metal atom complex. The \( \text{C}_{60}−\text{Co}_x \) spectrum exhibit the main features of the \( \text{C}_{60} \) single molecule suggesting that the \( \pi \) states are at most only slightly affected by the interaction between the molecule and the Co atoms. This is furtherly confirmed by the similarity between the conductance maps of \( \text{C}_{60} \) and \( \text{C}_{60}−\text{Co}_x \). The conductance map and the apparent height measurement suggest that the Co cluster is dissociated between the \( \text{C}_{60} \) molecule and the Au (111) surface. These experimental results were explained and supported by DFT electronic structure simulations.

**ASSOCIATED CONTENT**

**Supporting Information**

Conductance map, spin-polarized vacuum LDOS, and molecular dynamics simulations. This material is available free of charge via the Internet at http://pubs.acs.org.

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**Notes**

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